

# New Sidewall Materials in Aluminum Reduction Cell

Yawei Li, Yibiao Xu

Nowadays the heat dissipating has to be reduced as far as possible in new electrolysis technology in order to meet the needs of energy saving in aluminum industry. Ledge-free sidewall is preferred as it potentially reduces the energy requirement of aluminum production and allows the use of larger anode that increase capacity and productivity of a cell of the same dimensions. Also the environmental impacts would be reduced significantly in combination with inert anodes and cathode application. However, the sidewalls are facing the extreme challenge because they will be in direct contact with oxidizing, corrosive and reducing environments for different zones in aluminum cell. This article gives an overview of the current status and latest progress of such ledge-free sidewall materials. Also, an attempt to develop novel composite materials based on  $\text{MgO-NiFe}_2\text{O}_4\text{-TiO}_2$  and  $\text{MgO-SnO}_2\text{-TiO}_2$  system for ledge-free sidewall have been made in our laboratory. Finally, some advices on the development of the ledge-free sidewall materials in future are proposed.

## 1 Introduction

It is well-known that the Hall-Héroult process is only industrially viable one to produce aluminum. Such an electrolysis based process includes the dissolution of commercial alumina into molten electrolyte, and reduction into aluminum at about 970 °C using carbon anodes. During this process molten aluminum is generated at the cathode, and the carbon anode is oxidized and consumed to produce 70–90 %  $\text{CO}_2$  [1]. Meanwhile, in practical process,  $\text{Si}_3\text{N}_4$  bonded SiC refractory has been chosen as the sidewall materials in aluminum reduction cell, on inner side of which a frozen ledge of electrolyte forms because of its high thermal conductivity [2–5]. In this case a large heat accounting for approximately 35 % of the total input energy has to transfer through the sidewalls [6–8], which is the reason for the low energy efficiency of 40–45 % in current Hall-Héroult process.

Nowadays it come to realize that such heat dissipating should be reduced as far as possible in new electrolysis technology in order to meet the needs of energy saving in alu-

minum industry [9, 10]. For example, once a high insulation layer was applied outside the sidewalls in new technology, there was ledge-free on inner side and the heat in the cell could be kept rather than removed, leading to a potentially 30–40 % energy savings. Such a ledge-free sidewall may also allow the use of larger anodes to increase the capacity and productivity of the same dimension cell [11]. Furthermore, if the combination of inert anodes and cathode were used in new technology, the environmental impacts would be reduced significantly. However, in that case the sidewall would be exposed directly to the oxidizing gas, corrosive electrolyte and reducing aluminum from the top to bottom causing significantly the reduction of the sidewall service life if  $\text{Si}_3\text{N}_4$  bonded SiC materials were still used as sideling [12–14]. An alternative approach to tackling these difficulties is to develop a new material as sideling instead of the frozen ledge on the inner of the traditional sidewall [10]. However, the tough challenges to new sidewall material have to withstand all different sur-

rounding environments in aluminum reduction cells.

In present paper, the requirements and structure design for ledge-free sidewall will be introduced, and then the state of the art for sidewall materials is reviewed. Following that  $\text{MgO-NiFe}_2\text{O}_4\text{-TiO}_2$  and  $\text{MgO-SnO}_2\text{-TiO}_2$ , as candidate materials are presented on our research work. Finally, the development of new materials as sidewall is further proposed.

## 2 Requirements and structure design for ledge-free sidewall

The required properties of the ledge-free sidewall can be divided into physical, chemical as well as economical characteristics [16, 17], which are summarized as listed in Tab. 1. Based on the above criteria it is difficult to develop only a kind of material satisfying all the chemical requirements of the sidewall [9, 10], Mukhlis, et al. [9] have proposed multi-layer strategy and heat flux strategy to tackle the combining corrosive action on the different environments within the cell as illustrated in Fig. 1. Multi-layer strategy can be applied horizontally at gas, bath and metal zones where the different materials to be used can withstand each

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**Tab. 1** General selection criteria for an ideal ledge-free sidewall material

|                         |   |
|-------------------------|---|
| Physical requirements   | Low electrical conductivity                 |
|                         | Low thermal conductivity                    |
|                         | High mechanical strength                    |
|                         | High thermal shock resistance               |
|                         | Superior abrasion resistance to bath        |
|                         | Low porosity                                |
| Chemical requirements   | Not reactive to electrolyte and Al          |
|                         | Low solubility in molten electrolyte and Al |
|                         | High oxidation resistance                   |
|                         | Not wetted by molten electrolyte and Al     |
| Economical requirements | Low fabrication costs                       |
|                         | Ease of joining                             |

zone mentioned above (Fig. 1a). In general, the interface of gas and bath zone as well as bath and metal zone shifts during the electrolysis process. Therefore, the appropriate horizontal multi-layer strategy is to use two kinds of materials, namely, one of them can withstand the gas and bath zone while the other is in the bath and metal zone. Similarly, multi-layer strategy can be applied vertically that one kind of material with high corrosion resistance is within the cell while the other one with low heat conductivity to maintain the heat balance of the cell applied outside (Fig. 1b). This strategy would not be available because only a kind of material as inside wall could not meet the requirements for three different environments.

The third strategy is recommended in Fig. 1c, which controls heat flux through bath zone to allow ledge formation on this zone while heat flux through the air and metal zone is minimized by incorporating air gaps or inserting low heat conductivity material into sidewall. In this case a certain amount of energy has to be dissipated through the

sidewall, which make it difficult to achieve a potentially 30–40 % energy savings.

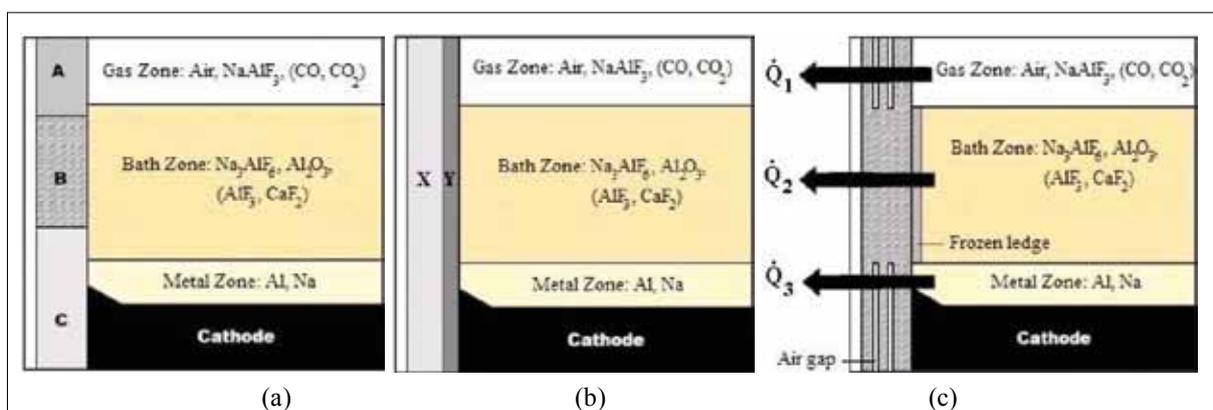
### 3 The state-of-the art for sidewall materials

Recently, the efforts have mainly focused on developing materials for both of the gas and bath zones where the sidewall is facing the most extreme challenge in these areas.  $\text{NiFe}_2\text{O}_4$  and  $\text{SnO}_2$  are in investigation for sidewall materials with respect to that these oxides have high stability in air and bath. In fact, these oxides are regarded as potential materials for inert anodes in aluminum reduction cell [16, 22, 23] which should possess the desirable properties similar to those of the sidewall materials except for high electrical conductivity.

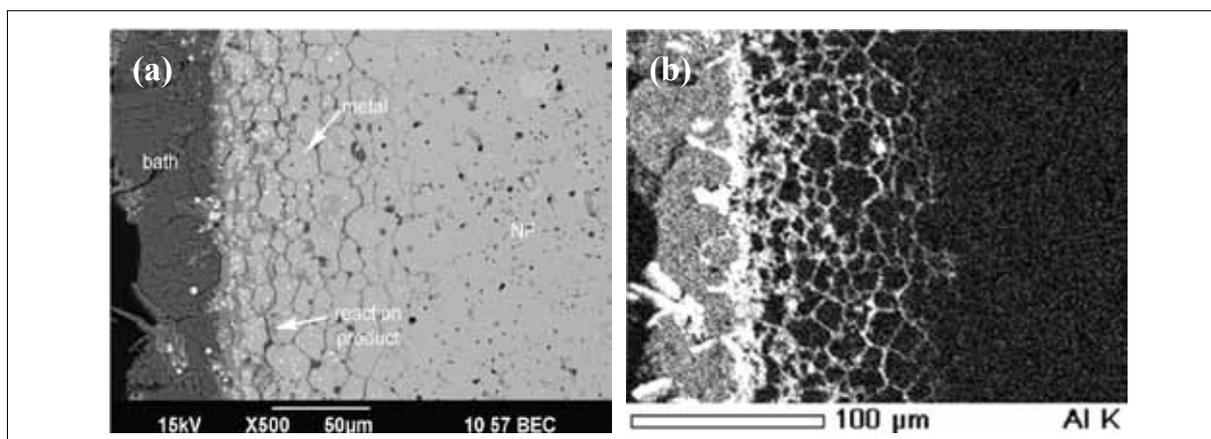
#### 3.1 Nickel ferrite as sidewall material

$\text{NiFe}_2\text{O}_4$  is of interest as novel sideling materials [11, 17, 22] due to its high resistance toward the electrolyte and oxygen [16, 19, 24], which could be used as a sidewall re-

fractory, either as a bulk refractory or as part of a multi-refractory system. Yan *et al.* [24] studied the dynamic corrosion of nickel ferrite samples in molten bath with specimen rotation speed of 25 rpm. Specimen with higher porosity was associated with more serious bath infiltration. Besides, the extent of corrosion of the nickel ferrite could be lowered by using high alumina content as well as low  $\text{NaF}/\text{AlF}_3$  molar ratios (CR). It was also proposed that the mechanism of corrosion involved grain boundary attack and therefore specimen with larger grain size had better corrosion resistance in the bath. Downie [22] has also confirmed that nickel ferrite specimen with a larger grain size showed better performance than that of specimen with smaller grains in cryolite-based baths under static conditions. Besides, aluminum-rich oxide phase formed inside the specimens with Fe-Ni metal alloy at the surface of the corroded part. Similarly, Sekhar, *et al.* [25] observed the formation of a metallic phase by the immersion of nickel ferrite in molten cryolite. Nightingale, *et al.* [11] reported some larger Fe-Ni metallic particles in the residual bath and many smaller ones are scattered throughout the reaction zone after nickel ferrite was tested in cryolite bath containing 10 % alumina for 4 h (Fig. 2a). Grain boundaries were being attacked accompanying with the formation of aluminum rich phase in the grain boundaries (Fig. 2b), and grains separated from the bulk. And it suggested that additives in nickel ferrite spinel based materials could impart resistance to grain boundary penetration in combination with increasing the grain size. In addition, Olsen and Thonstad [26, 27] observed that a dense layer of about 50  $\mu\text{m}$  formed at the surface



**Fig. 1a–c** Horizontal layer strategy (a), vertical layer strategy (b), and heat flux strategy (c)



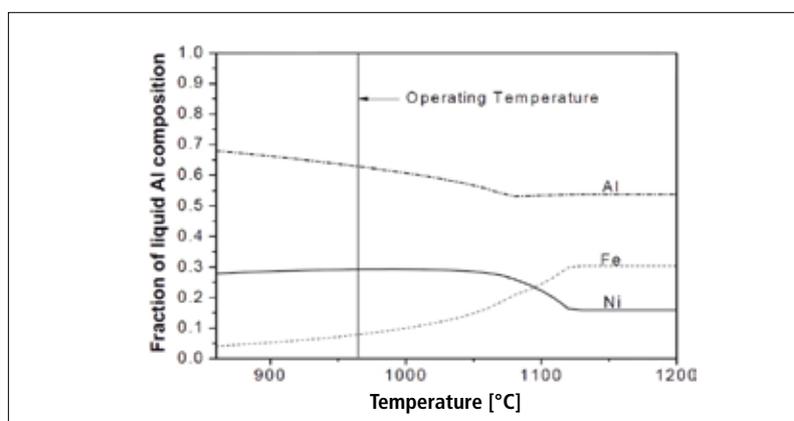
**Fig. 2a–b** Nickel ferrite after corrosion in 10 % alumina-cryolite bath at 980 °C for 4 h (a), and elemental map of Al for the same region (b)

of nickel ferrite based anodes after corroded in alumina-cryolite bath, which were composed of aluminates, such as  $\text{FeAl}_2\text{O}_4$  and  $\text{NiAl}_2\text{O}_4$ , hindering further chemical corrosion in the molten electrolyte.

Overall, the extent of corrosion of the nickel ferrite in all the experiments above is quite small. Thus, nickel ferrite might be suitable to be applied in the gas zone and bath zone and unfit in the metal zone due to the dissolution of iron and nickel to aluminum as shown in Fig. 3 [10, 28].

### 3.2 $\text{SnO}_2$ as sidewall material

$\text{SnO}_2$  also demonstrates extremely low solubility in cryolite melt under oxidizing atmosphere [29–35], which may be another candidate material for sidelining. *Xiao et al.* [29] reported that the solubility of  $\text{SnO}_2$  increased with decreasing  $\text{Al}_2\text{O}_3$  concentration and increased with increasing temperature, and it showed a maximum at  $\text{NaF}/\text{AlF}_3$  molar ratio (CR) of 3. Meantime, the solubility of  $\text{SnO}_2$  was associated with oxygen partial pressure over the melts due to formation of divalent tin ( $\text{Sn(II)}$ ) species. *Xiao, et al.* [30] also confirmed that high current densities led to serious corrosion of the anode specimens as well as low  $\text{NaF}/\text{AlF}_3$  molar ratios and low alumina concentrations. *Keller, et al.* [32] pointed out that the electrolyte corrosion mechanism was mainly governed by solubility of  $\text{SnO}_2$  ceramic and the mass transport condition. With respect to additives, *Vecchio-Sadus, et al.* [36] examined the performance of  $\text{SnO}_2$  ceramic anodes doped with copper and antimony oxides in cryolite alumina melts.  $\text{CuO}$  additive could enhance the sinterability of the  $\text{SnO}_2$  based ceramic [39].



**Fig. 3** Equilibrium calculations of 1 mol nickel ferrite and 10 mol aluminum at 860 – 1200 °C

However, after corrosion test, the Cu depletion zone appeared causing a subsequent decrease in mechanical strength of the ceramic [34, 35].

It is worth mentioning that though the pure  $\text{SnO}_2$  materials obtained high chemical stability in molten electrolyte under the oxidizing atmosphere, they still can not be applied in the aluminum zone since it can be easily reduced into metal Sn [37].

### 3.3 Aluminate spinel as sidewall material

*Jentoftsen, et al.* reported some aluminate spinels, such as  $\text{FeAl}_2\text{O}_4$  and  $\text{NiAl}_2\text{O}_4$ , also possess high chemical stability in the cryolite-alumina melts [38]. And *Olsen and Thonstad* [26] confirmed that the corrosion rate of nickel ferrite anodes in cryolite bath was reduced obviously when an aluminates spinels layer formed at the surfaces. Recently, *Yan, et al.* [10] prepared a series of aluminate spinels, including  $\text{NiAl}_2\text{O}_4$ ,  $\text{MgAl}_2\text{O}_4$  and  $\text{Ni}_x\text{Mg}_{(1-x)}\text{Al}_2\text{O}_4$  solid solutions, as candidate materials for the sidewalls. It revealed

that the  $\text{NiAl}_2\text{O}_4$  and  $\text{MgAl}_2\text{O}_4$  spinels had good corrosion resistance toward the melts, with solubility of Ni from  $\text{NiAl}_2\text{O}_4$  being 0,01 mass-% and Mg from  $\text{MgAl}_2\text{O}_4$  being 0,07 mass-%. Interestingly, it was also found that the concentrations of Ni and Mg from the  $\text{Ni}_{0,5}\text{Mg}_{0,5}\text{Al}_2\text{O}_4$  solid solution in the bath were lowered to 0,007 mass-% Ni and 0,05 mass-% Mg due to reduced activities of  $\text{NiO}$  and  $\text{MgO}$  in the spinel solid solution. It is deduced that the  $\text{Ni}_{0,5}\text{Mg}_{0,5}\text{Al}_2\text{O}_4$  solid solution might be also more difficult to reduce by liquid Al and proposed that  $(\text{Ni,Mg})\text{Al}_2\text{O}_4$  solid solutions could be potentially promising materials for the ledge-free sidewalls.

### 4 Magnesia-based composite as sidewall material

In fact, magnesia based compositions could be candidates applied in the gas and bath zone due to its good resistance to basic slags as well as low price [40]. The magnesia materials would not contaminate Al melt since the reaction products of  $\text{MgO}$  and

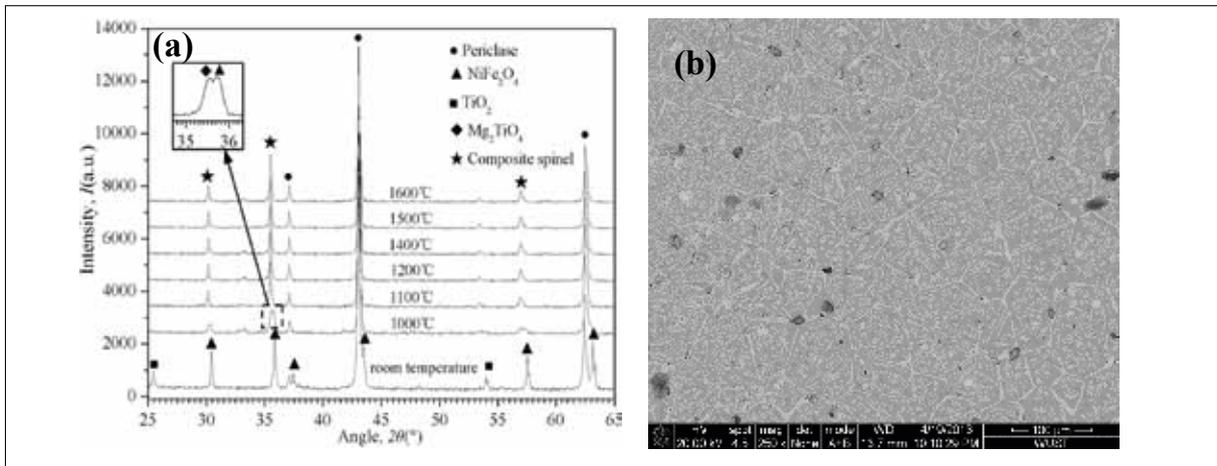


Fig. 4a–b XRD patterns of specimen MTN30 at various temperatures (a), and its micrograph fired at 1600 °C (b)

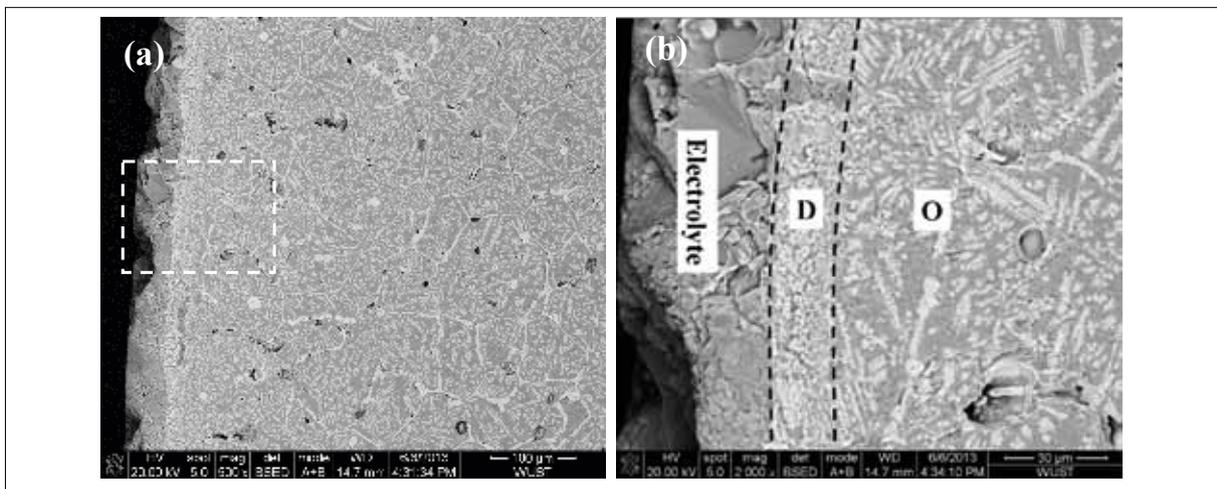


Fig. 5a–b SEM images of the corroded surface of specimen MTN30

electrolyte were mainly  $MgF_2$  and  $Al_2O_3$  [18]. In order to increase the corrosion resistance, high chemical substance  $NiFe_2O_4$  or  $SnO_2$  can be added into the composite. Moreover,  $TiO_2$  can enhance densification magnesia materials [42], slowing down the penetration of electrolyte [24, 43]. Based on the above consideration, novel composites based on  $MgO-SnO_2-TiO_2$  and  $MgO-NiFe_2O_4-TiO_2$  systems for ledge-free sidewall were prepared and some experimental results were presented to demonstrate their sinterability and chemical stabilities in molten electrolyte in air.

#### 4.1 $MgO-NiFe_2O_4-TiO_2$ system

The composition of the  $MgO-NiFe_2O_4-TiO_2$  materials (labeled as MTN30) was 65 mass-%  $MgO$ , 30 mass-%  $NiFe_2O_4$  and 5 mass %  $TiO_2$ . The XRD patterns (Fig. 4a) indicated that reaction between  $MgO$  and  $TiO_2$  occurred to generate  $Mg_2TiO_4$  that

formed solid solution with  $NiFe_2O_4$  to generate composite spinel ( $Ni_xTi_{1-x}Fe_{2x}Mg_{2-2x}O_4$ ) during sintering process, which promoted mass transfer and enhanced densification of the materials simultaneously (Fig. 4b). And light grey composite spinel distributed homogeneously in the matrix of the specimen.

The corroded SEM micrographs of MTN30 (Fig. 5a) revealed that no obvious corrosion phenomenon can be observed after 10 h immersion in electrolyte (57,8 mass-%  $K_3AlF_6$ -16,7 mass-%  $Na_3AlF_6$ -25,5 mass-%  $AlF_3$ ) at 900 °C. And the dense layer (D) of 20  $\mu m$  formed (Fig. 5b) that was identified as solid-state solution of aluminates and  $NiFe_2O_4$  by EDS analysis. The layer functions as the blocking layer between the origin layer (O) and electrolyte to improve the corrosion resistance [44].

Fig. 6 reveals phase evolution of 100 g MTN30 in electrolyte calculated by thermo-

dynamic analysis software (FactSage 6.2). It can be seen that with increasing electrolyte content,  $MgO$  decreases continuously accompanying with the formation of fluorides and  $MgAl_2O_4$  while no obvious changes can be found on the contents of  $NiFe_2O_4$  and  $Mg_2TiO_4$ .

It indicated that  $NiFe_2O_4$ ,  $Mg_2TiO_4$  and their solid-state solution (the composite spinel) possessed better chemical stability than  $MgO$  in molten electrolyte.

#### 4.2 $MgO-SnO_2-TiO_2$ system

The composition of the  $MgO-SnO_2-TiO_2$  materials (labeled as MTS10) was 85 mass-%  $MgO$ , 10 mass-%  $SnO_2$  and 5 mass-%  $TiO_2$ . XRD pattern of MTS10 (Fig. 7a) revealed that  $Mg_3SnO_4$  and  $Mg_2TiO_4$  formed and then solid solution  $Mg_2Ti_{1-x}Sn_xO_4$  during the heating process, which led to mass transfer and densification of the materials [45] (Fig. 7b).

The corroded SEM micrographs of MTS10 (Fig. 8a) presented that only a corrosion layer (C) of 30  $\mu\text{m}$  formed at the surface of the specimen after 10 h test at 900  $^{\circ}\text{C}$ . And the elemental map of Al in Fig. 8b revealed that the infiltration of electrolyte was completely inhibited. It indicated that the MTS10 also demonstrated high corrosion resistance to the molten electrolyte, which was attributed to the lowering infiltration of electrolyte as well as high chemical stable composite spinel retarding chemical corrosion.

### 5 Outlook on the development of ledge-free sidewalls

The development of ledge-free sidewalls brings great economical and environmental opportunities in aluminum industry. Meantime, the sidewall materials are facing the critical challenge in service. The following issues of the research works need to be taken into consideration:

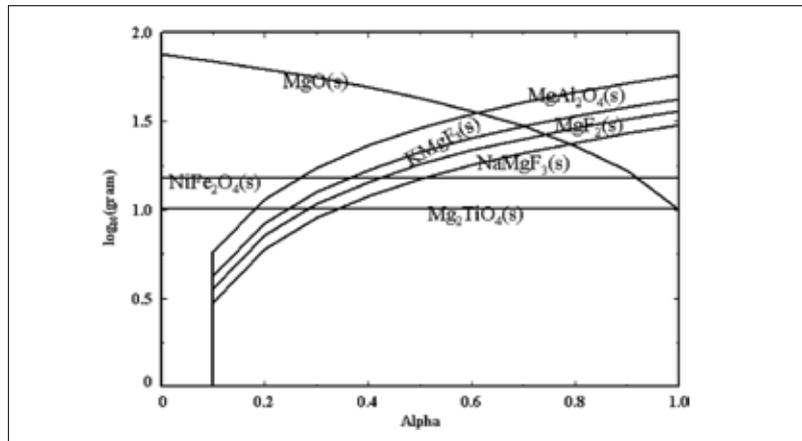


Fig. 6 predicted phase evolution of specimen MTN30 in the electrolyte at 900  $^{\circ}\text{C}$

- (1) New design for ledge-free sidewalls in aluminum cell is in need for selection criteria of sideling since a single material can hardly satisfy all the chemical requirements. Horizontal multi-layer strategy with two different materials is recommended, in which one of them can withstand the gas and bath zones while the other can withstand the bath and metal zones.
- (2)  $\text{NiFe}_2\text{O}_4$  and  $\text{SnO}_2$  materials, candidates for inert anodes in aluminum reduction

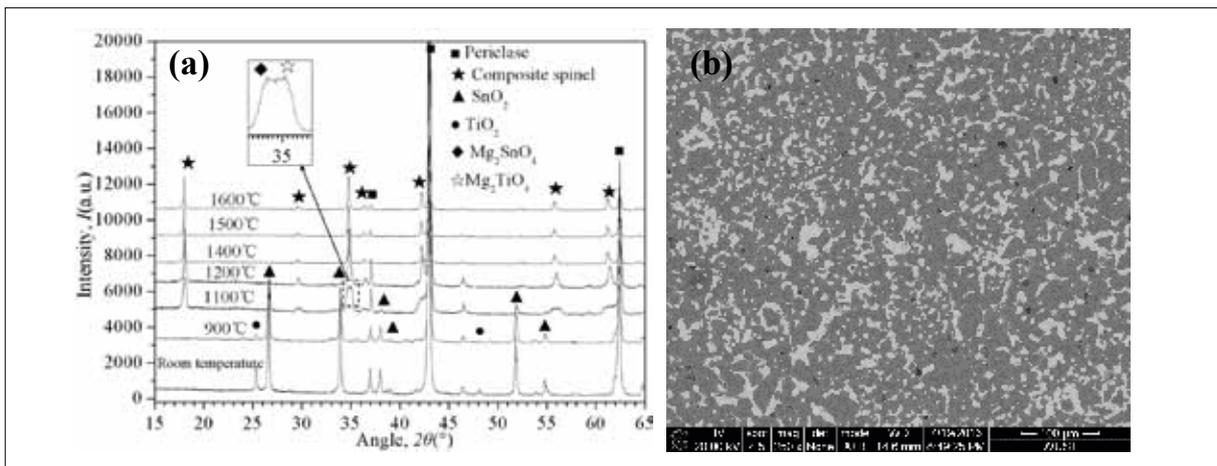


Fig. 7a–b XRD patterns of specimen MTS10 (a), and micrograph of specimen MTS10 fired at 1600  $^{\circ}\text{C}$  (b)

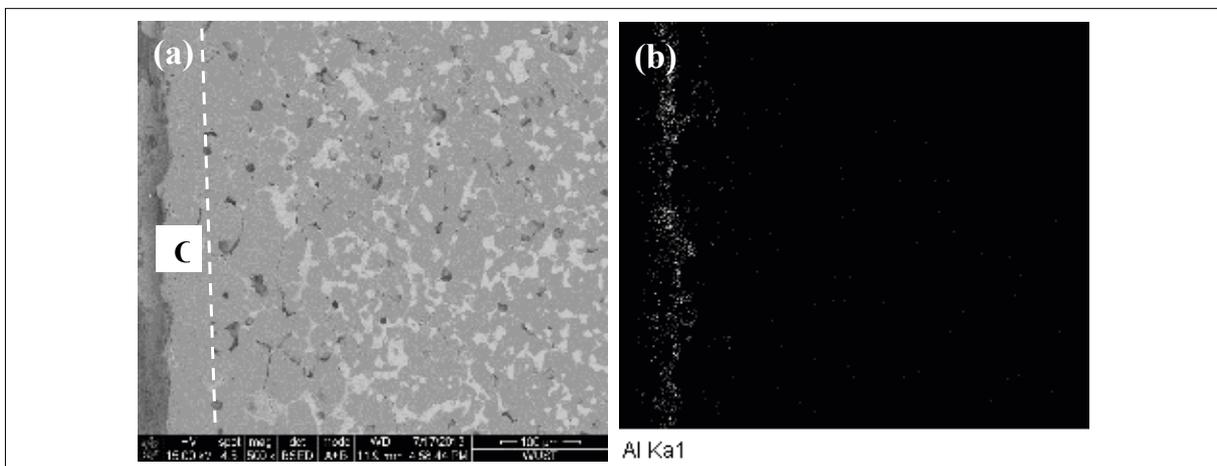


Fig. 8a–b SEM images of corroded specimen MTS10 (a), and elemental map of Al for the same region (b)

cell are also potential for the ledge-free sidewall in gas and bath zones. Furthermore, some aluminate spinel and composites containing high stability substances, e.g.  $\text{NiFe}_2\text{O}_4$  and  $\text{SnO}_2$ , also deserve special attention since the price of pure  $\text{NiFe}_2\text{O}_4$  and  $\text{SnO}_2$  is too high to be used as sidewall materials.

- (3) Issues relating to joining of materials, process control, manufacturing and maintenance also need to be addressed before new sidewall designs can be implemented.

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